

## Revised Structure of the Orthoborate YBO<sub>3</sub>

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The YBO<sub>3</sub> orthoborate was prepared by a flux evaporation process and the diffraction lines of its X-ray pattern were indexed on the basis of single crystal data. Crystal structure was determined ( $R=0.035$ ) in the  $P6_3/m$  space group from data recorded on a four-circle automatic diffractometer, with cell parameters  $a=3.776$  (1) Å,  $c=8.806$  (4) Å. The structure consists of a three-dimensional network made up of eightfold coordinated yttrium atoms and fourfold coordinated boron atoms. The structure exhibits two nonequivalent environments for the yttrium ions, which is confirmed by luminescence studies using the Eu<sup>3+</sup> ion as structural probe. This work was completed by <sup>11</sup>B NMR and IR studies, which show that boron has tetrahedral coordination.

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### 1. INTRODUCTION

During the past few years, much interest has been devoted to the study of yttrium and lanthanide orthoborates because of their high UV transparency and exceptional optical damage threshold, which make them attractive for numerous practical applications.

Nevertheless, the large number of studies concerning the rare earth orthoborates LnBO<sub>3</sub> have not always clearly identified their crystallographic structures.

In 1961, Levin *et al.* (1) investigated their polymorphism and reported on the regions of stability of the different structures in relation with borates preparation temperature and the ionic radius of the rare earth. The diagram they established showed that the structures of the rare earth borates are related to the three crystalline forms of CaCO<sub>3</sub>, aragonite, vaterite, and calcite, depending on the rare earth.

The compounds LaBO<sub>3</sub>, NdBO<sub>3</sub> (1), and CeBO<sub>3</sub> (2) exhibit the aragonite-type structure up to 1450, 1300, 1500°C, respectively, and crystallize in the orthorhombic system with space group  $Pnam$  ( $Z=4$ ). A recent study of the luminescence of Eu<sup>3+</sup> in NdBO<sub>3</sub> (3) confirms these results

and proposes a  $C_3$  symmetry site for the rare earth. At higher temperatures, two different forms (denoted H) have been found for LaBO<sub>3</sub> and NdBO<sub>3</sub>. According to Bohlhoff *et al.* (4), they crystallize in the monoclinic system. Moreover, we have pointed out that these H-forms also exist at low temperature (about 600°C) when the compounds are prepared by the sol-gel technique.

For the smallest rare earth ions (Sm to Lu) the orthoborates possess the vaterite structure, which is consistent with a pseudo-hexagonal symmetry. In this series, SmBO<sub>3</sub> and EuBO<sub>3</sub> present both vaterite and H-forms. However, there remains some ambiguity. For YBO<sub>3</sub>, Newnham *et al.* (5) proposed two space groups, a disordered hexagonal  $P6_3/mmc$  with only one  $D_{3d}$  point symmetry for the rare earth, and an ordered hexagonal  $P6_3/mcm$ , with two kinds of point symmetry sites for the rare earth ( $D_3$  and  $D_{3d}$ ). Then Bradley (6) confirmed the  $D_3$  point symmetry for the rare earth but using the  $P6c2$  space group. Despite the uncertainty about the crystal structure of YBO<sub>3</sub>, several studies using IR, NMR, and Raman data pointed out that the boron atoms form a B<sub>3</sub>O<sub>9</sub> unit with tetrahedral coordination (7–9). Recently, Hölsä (10), studying the luminescence of Eu<sup>3+</sup> as a structural probe in LuBO<sub>3</sub>, proposed for the vaterite type two kinds of sites for Lu, with  $D_{3d}$  and  $T$  point symmetries.

In the third series of borates, InBO<sub>3</sub>, LuBO<sub>3</sub>, and ScBO<sub>3</sub> (1, 11–13) are known to exhibit the calcite type structure. Let us note that LuBO<sub>3</sub> presents both a low-temperature calcite form and a high-temperature vaterite form. The calcite structure has a rhombohedral symmetry. The space group is  $R\bar{3}c$  ( $Z=6$ ) with a single site for the rare earth having  $S_6$  point symmetry.

For the vaterite structure which is our aim the luminescence results given in the literature lead to two kinds of sites,  $D_{3d}$  and  $T$ , for the rare earth, as mentioned before (10). Our optical studies on YBO<sub>3</sub>:Eu<sup>3+</sup> have led to two intrinsic sites of  $C_3$  symmetry (14). These data are not consistent with the crystallographic results (10, 14). Because of these inconsistencies, in this paper, we report the revised crystal structure of YBO<sub>3</sub> determined from X-ray single crystal data. This structure is compared with that of InBO<sub>3</sub>.

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## EXPERIMENTAL

## Synthesis

Powder sample of  $\text{YBO}_3$  was prepared by heating in air a mixture of the reagents  $\text{Y}_2\text{O}_3$  (Rhône Poulenc 99.99%) and  $\text{H}_3\text{BO}_3$  (Prolabo 99%) in the ratio  $\text{Y/B} = 1.2$ . The mixture was heated in a Pt crucible at  $500^\circ\text{C}$  for 2 hr, ground, and again heated at  $1100^\circ\text{C}$  for 2 hr. The flux evaporation process using  $\text{LiBO}_2$  (Strem Chemicals 99.9%) at a solvent has been used for the synthesis of single crystals of  $\text{YBO}_3$ . The procedure was as follows (11):

$\text{YBO}_3$  (25% wt) and  $\text{LiBO}_2$  (75% wt) were intimately mixed together on a vibrating machine and then put into a platinum crucible with a closely fitting lid limiting vaporization of the solvent. The crystal growth was carried out in a furnace, the temperature of which was regulated and programmed by a 9 P Eurotherm controller. After the crucible was introduced into the furnace, the thermal program was initiated. The starting materials were heated up to  $1150^\circ\text{C}$  at a rate of  $100^\circ\text{C/h}$  then maintained for 10 hr at this temperature and cooled to  $840^\circ\text{C}$  at a rate of  $3^\circ\text{C/h}$  and finally to room temperature at a rate of  $100^\circ\text{C/h}$ . Owing to the acidic character of  $\text{YBO}_3$ , a basic flux is required for dissolution and growth (15).  $\text{LiBO}_2$  was chosen both for its basic properties and for its low melting point. After washing with hot distilled water  $\text{LiBO}_2$  was dissolved and crystals were obtained.

## Structure Determination

A colorless crystal was selected from the synthesized specimen. Unit cell dimensions were determined accurately using a four-circle Nonius CAD4 diffractometer by centering on 25 reflections with  $14 \leq 2\theta \leq 40$  and least-squares refinement of the measured setting angles. Data collection parameters are given in Table 1.

At first, the superstructure with  $a' = a\sqrt{3}$  proposed by Bradley does not prove necessary to index the X-ray diffraction pattern. Moreover, further refinements were not successful.

The crystal structure of  $\text{YBO}_3$  was solved in the  $P6_3/m$  space group (No. 176) with the program MolEN (16), the atomic scattering and anomalous dispersion factors for atoms being taken from the International Tables for X-Ray Crystallography (17). The refinement converged to  $R = 0.035$  after introduction of anisotropic temperature factors except for boron atom; a secondary extinction correction improved the refinement. Atomic coordinates with isotropic thermal parameters are listed in Table 2, anisotropic displacement parameters in Table 3, and significant interatomic distances and angles in Table 4.

The powder X-ray diffraction patterns are indexed with cell parameters  $a = 3.772 \text{ \AA}$ ,  $c = 8.791 \text{ \AA}$ , which are in agreement with the single crystal results (Table 1). The

TABLE 1  
Crystal Data and Details of the X-Ray Data Collection of  $\text{YBO}_3$

Chemical formula	$\text{YBO}_3$
Formula weight	147.71
Space group	$P6_3/m$ (No. 176)
Cell dimensions	$a = 3.776(1) \text{ \AA}$ $c = 8.806(4) \text{ \AA}$
Z	2
Density (calculated)	$2.26 \text{ g/cm}^3$
Linear absorption coefficient	$133.4 \text{ cm}^{-1}$
Crystal size (mm)	$0.03 \times 0.18 \times 0.35$
Radiation	$\text{MoK}\alpha(0.71073) \text{ \AA}$ graphite monochromated
Temperature	293 K
Scan:	
mode	$\omega-2\theta$
width	$(0.8 + 0.35 \tan \theta)^\circ$
aperture	$(2.7 + 0.4 \tan \theta) \text{ mm}$
absorption correction	empirical DIFABS [22]
$T_{\min}/T_{\max}$	0.82/1.15
Data collected	$0 \leq \theta \leq 45$ $0 \leq h \leq 7$ $0 \leq k \leq 7$ $0 \leq l \leq 17$
Measured reflections	324
Independent reflections ( $I > 3\sigma(I)$ )	182
Number of refined parameters	16
Extinction coefficient	$2.138 \times 10^{-6}$
Weighting scheme	unit
Final Fourier residuals	$1.67 e \text{ \AA}^{-3}$ , and $-1.12 e \text{ \AA}^{-3}$
Final R indices ( $I > 3\sigma(I)$ )/S	(0.035–0.041)/1.066
R indices (all data)	0.080–0.077

observed and calculated  $d$  spacings and the associated relative intensities of the powder X-ray diffraction reflections are listed in Table 5.

Let us mention that comparison between theoretical X-ray diffraction pattern ( $P6_3/m$  space group) and observed X-ray diffraction pattern of  $\text{YBO}_3$  provides evidence for the new crystallographic results which it is not the case for the assumption of Bradley.

TABLE 2  
Final Atomic Coordinates and Equivalent Isotropic Thermal Parameters for  $\text{YBO}_3$

Atoms	Wyckoff position	Sites occupancy	x	y	z	$B_{\text{eq}} (\text{Å}^2)^a$
Y	2 (b)	1	0	0	0	0.273(6)
O(1)	4 (f)	1	0.667	0.333	0.1103(7)	1.05(6)
O(2)	6 (h)	1/3	0.774(4)	-0.119(4)	0.25	0.5(2)
B	6 (h)	1/3	0.586(5)	0.435(5)	0.25	0.4(2) <sup>b</sup>

<sup>a</sup>  $B_{\text{eq}} = (4/3)[a^2B(1, 1) + b^2B(2, 2) + c^2B(3, 3) + ab(\cos \gamma)B(1, 2) + ac(\cos \beta)B(1, 3) + bc(\cos \alpha)B(2, 3)]$ .

<sup>b</sup> Atom refined isotropically.

**TABLE 3**  
General Displacement Parameters

	B(1.1)	B(2.2)	B(3.3)	B(1.2)	B(1.3)	B(2.3)
Y	0.41(1)	0.41(1)	0.00(1)	0.41(1)	0	0
O(1)	1.2(1)	1.2(1)	0.7(2)	1.2(1)	0	0
O(2)	0.4(3)	0.2(2)	0.7(3)	-0.1(2)	0	0

Eu<sup>3+</sup> doped orthoborate single crystals have been prepared in the same way as described before by replacing molar proportion of Y<sub>2</sub>O<sub>3</sub> by Eu<sub>2</sub>O<sub>3</sub> (Rhône-Poulenc 99.99%). In all cases, the structure refinement shows that these compounds are isostructural with YBO<sub>3</sub>.

### DESCRIPTION OF THE STRUCTURE

Figure 1 represents the projection of YBO<sub>3</sub> unit cell content onto the (001) plane. Y is eightfold coordinated by oxygen atoms in an arrangement which can be described as a trigonal bipyramidal antiprism. These oxygen surrounding the Y<sup>3+</sup> ions occupy two positions in the cell: six O(1) atoms in 4 (f) and two O(2) atoms in 6 (h) with a partial occupancy (1/3). The Y–O(1) and Y–O(2) distances are respectively 2.39 and 2.32 Å. Yttrium environments are presented in Fig. 2. Other distributions of O(2) in 6 (h) sites are not possible for steric reasons.

Boron is fourfold coordinated; [BO<sub>4</sub>] tetrahedra are in two forms, the first one (I) very regular with B–O distances ranging from 1.37 to 1.57 Å and the second one (II) with two similar bonds (1.37 Å) and two larger ones between 1.89 and 1.92 Å. [YO<sub>8</sub>] polyhedra share edges O(1)–O(1) and O(2) corners with two other [YO<sub>8</sub>] polyhedra. A view of the crystal structure of YBO<sub>3</sub> along c axis is shown in Fig. 3a. The structure can be described as a 3D unit by [YO<sub>8</sub>]

**TABLE 4**  
Selected Interatomic Distances and Angles for YBO<sub>3</sub>

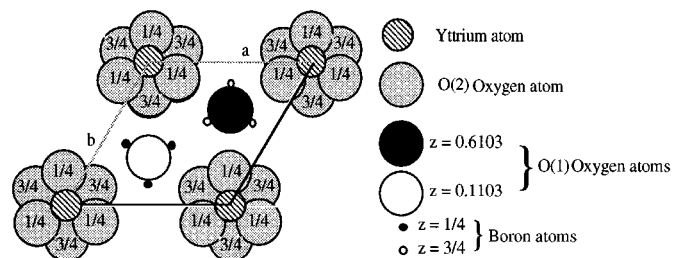
Y environment					
Y–O(1)	2.386 (2) × 6	O(1)–Y–O(1)	75.4 (1)	O(1)–Y–O(1)	180.0 (1)
Y–O(2)	2.323 (4) × 2	O(2)–Y–O(2)	180.0 (1)	O(2)–Y–O(2)	147.9 (4)
⟨Y–O⟩	2.37	O(1)–Y–O(2)	58.4 (4)	O(1)–Y–O(2)	84.6 (3)
		O(1)–Y–O(2)	57.4 (3)		
B environment					
I	II				
B–O(1)	1.37 (1) × 2	B–O(1)	1.37 (1) × 2	O(1)–B–O(1)	128.0 (2)
B–O(2)	1.47 (2) × 1	B–O(2)	1.92 (1) × 1	O(1)–B–O(2)	105.9 (6)
B–O(2)	1.57 (1) × 1	B–O(2)	1.89 (1) × 1	O(1)–B–O(2)	111.0 (2)
⟨B–O⟩	1.44	⟨B–O⟩	1.64	O(1)–B–O(2)	102.8 (7)

**TABLE 5**  
X-Ray Powder Pattern of YBO<sub>3</sub>

$d_{\text{exp}}$ (Å)	$d_{\text{calc}}$ (Å)	$h$	$k$	$l$	$I/I_0$
4.4153	4.3956	0	0	2	64.6
3.2752	3.2668	1	0	0	100
3.0710	3.0622	1	0	1	13.7
2.6283	2.6220	1	0	2	83.0
2.2017	2.1978	0	0	4	12.0
2.1843	2.1814	1	0	3	4.6
1.8896	1.8861	1	1	0	33.7
1.8268	1.8235	1	0	4	36.5
1.7364	1.7333	1	1	2	23.6
1.6325	1.6334	2	0	0	7.6
1.6090	1.6059	2	0	1	2.1
1.5333	1.5311	2	0	2	11.6
1.4678	1.4652	0	0	6	1.8
1.4312	1.4313	1	1	4	5.1
1.3384	1.3369	1	0	6	7.5
1.3134	1.3110	2	0	4	6.4
1.2364	1.2347	2	1	0	4.4
1.1899	1.1887	2	1	2	8.1
1.1574	1.1571	1	1	6	2.6

polyhedra, in which each [BO<sub>4</sub>] group is connected with two adjacent [BO<sub>4</sub>] groups by O(2) atoms and with two adjacent [YO<sub>8</sub>] groups by O(1) atoms. The projection of the structure on the [110] orientation of YBO<sub>3</sub> presented in Fig. 3a is compared with InBO<sub>3</sub> in the same orientation (Fig. 3b). The geometry of the [YO<sub>8</sub>] framework is not similar to that observed for [InO<sub>6</sub>] in InBO<sub>3</sub> (12). The important difference between the two structures deals with the existence of two more oxygens in the Y polyhedra (6 O(1) and 2 O(2)).

Considering the nature of the Y site in YBO<sub>3</sub>, it appears that Y atoms present a unique coordination but two different environments can be observed because of O(2) statistical distribution. Deviation from the ideal local S<sub>6</sub> symmetry is due to the existence of these two O(2) atoms in comparison with the S<sub>6</sub> symmetry observed for In in InBO<sub>3</sub>.



**FIG. 1.** Projection of the structure of YBO<sub>3</sub> on the (001) plane.

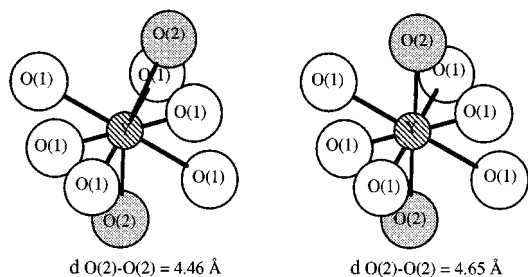


FIG. 2. Yttrium environment in  $\text{YBO}_3$ .

### SPECTROSCOPIC RESULTS

Infrared spectroscopy and  $^{11}\text{B}$  nuclear magnetic resonance studies in crystalline borates were carried out in order to specify the coordination of boron atoms.

IR absorption spectra were recorded with a Perkin-Elmer 16 PC spectrophotometer in the  $4000\text{--}400\text{ cm}^{-1}$  wavenumber range using KBr pellets. Solid state NMR spectra were recorded with a Bruker DSX 400 spectrometer, operating at the  $^{11}\text{B}$  resonance frequency of  $128,389\text{ MHz}$  with  $(\text{C}_2\text{H}_5)\text{OBF}_3$  as external standard. Magical angle spinning technique was used, with a rotation frequency of  $12\text{ KHz}$ .

### Infra-Red Spectroscopy

Figure 4 shows the spectra of  $\text{LnBO}_3$  ( $\text{Ln} = \text{La}, \text{Y}, \text{In}$ ) samples prepared by the usual solid state reaction as described previously (14). Vibrations frequencies are listed in Table 6.

In the  $1100\text{--}850\text{ cm}^{-1}$  range vibrations of B-O groups are certainly observed. For borates with calcite and

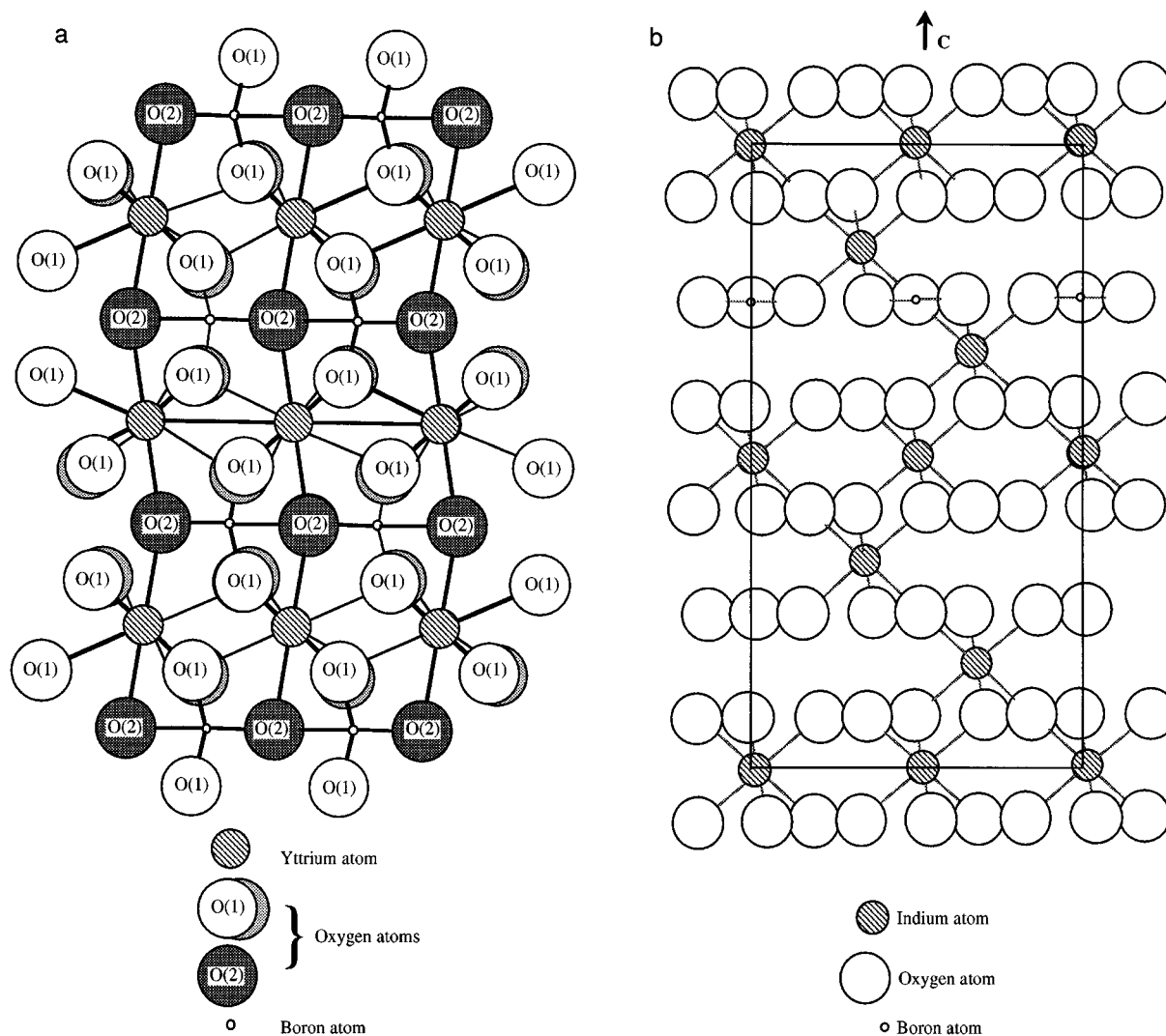


FIG. 3. (a) Projection of the structure of  $\text{YBO}_3$  with a  $[110]$  orientation. For the representation O(2) atoms have been arbitrarily placed in the sites corresponding to the second arrangement of Fig. 2. (b) Ball and stick representation of the  $\text{InBO}_3$  unit cell with a  $[110]$  orientation.

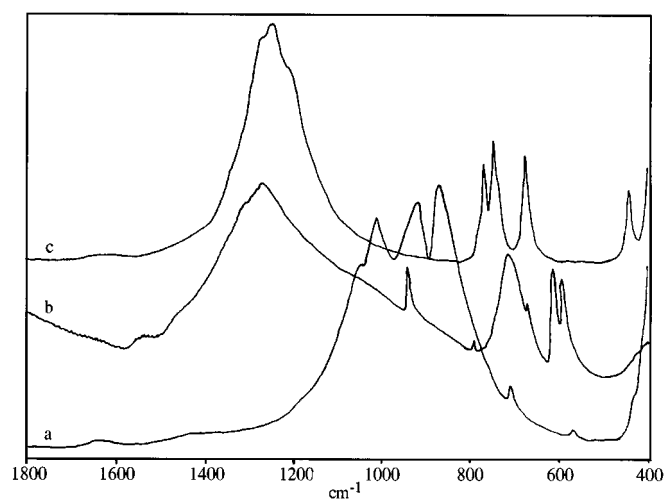


FIG. 4. Infrared spectra of (a) YBO<sub>3</sub>, (b) LaBO<sub>3</sub>, and (c) InBO<sub>3</sub>; prepared by dry method.

aragonite structures (18–19), the boron atoms are threefold coordinated. These results agree with the crystallographic studies. Both aragonite and calcite type borates have vibration modes as follows:  $\nu_3$  (asymmetric stretching) near

TABLE 6  
Observed Frequencies of the Different Vibration Modes  
in the YBO<sub>3</sub>, LaBO<sub>3</sub>, and InBO<sub>3</sub> Phases

InBO <sub>3</sub> ( $\nu$ cm <sup>-1</sup> ) (calcite)	LaBO <sub>3</sub> ( $\nu$ cm <sup>-1</sup> ) (aragonite)	YBO <sub>3</sub> ( $\nu$ cm <sup>-1</sup> ) (pseudo-vatérite)
	1541	
1269 $\nu_3$	1272 $\nu_3$	
1245 $\nu_3$		
1205 $\nu_3$		
	1089 $\nu_3$	
		1051
		1011
	938 $\nu_1$	919
		871
		844
	793 $\nu_2$	
764 $\nu_2$		
740 $\nu_2$		
	712 $\nu_2$	
671 $\nu_4$		709
	607 $\nu_4$	
	585 $\nu_4$	
		569
	529	
440		

1300 cm<sup>-1</sup>,  $\nu_2$  (out-of-plane bending) near 740 cm<sup>-1</sup>, and  $\nu_4$  (in plane bending) in the 670–500 cm<sup>-1</sup> region. Furthermore, aragonite type borates, in accordance with the selection rules for C<sub>s</sub> site symmetry, present  $\nu_1$  (symmetric stretching) near 940 cm<sup>-1</sup>. The frequency of B–O vibrations depends on boron coordination, i.e., tetrahedral or trigonal (7). The stretching frequencies of a coordinated XO<sub>n</sub> group decrease as coordination number  $n$  increases. Therefore, according to the observed frequencies in this range for YBO<sub>3</sub>-type phases, a coordination number of 4 for boron is more probable (9).

The spectra of both calcite and aragonite structures present a relatively intense band in the 700–800 cm<sup>-1</sup> range. This vibration corresponds to the  $\nu_2$  deformation vibrations of BO<sub>3</sub> trigonal groups. This band was not observed in borates having pseudo-vatérite structure. Therefore, these results are in good agreement with the crystallographic study.

### NMR

The coordination of boron atoms in boron compounds may be correlated to the nuclear magnetic resonance (NMR) signals arising from boron nuclei. A study of the solid state <sup>11</sup>B NMR spectra for YBO<sub>3</sub> and LaBO<sub>3</sub> has been undertaken in order to distinguish boron atoms in threefold and fourfold coordination and to confirm the conclusion based on X-ray and IR studies.

The NMR of LaBO<sub>3</sub>:Eu<sup>3+</sup> spectra are easy to analyze (Fig. 5). The two groups of lines (16.7–6.4 and 10.8–3.4 ppm) observed on the spectrum are characteristic of trigonal borons located in two different environments (20, 21). The small peaks which appear around 47.2, 38.0, and –14.6 ppm for example correspond to rotation bands. The comparison between the YBO<sub>3</sub>:Eu<sup>3+</sup> and LaBO<sub>3</sub>:Eu<sup>3+</sup>

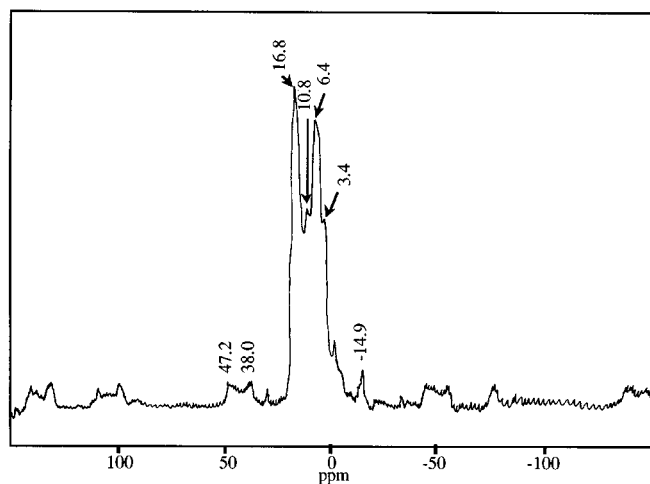


FIG. 5. <sup>11</sup>B NMR spectra of LaBO<sub>3</sub> prepared by dry method.

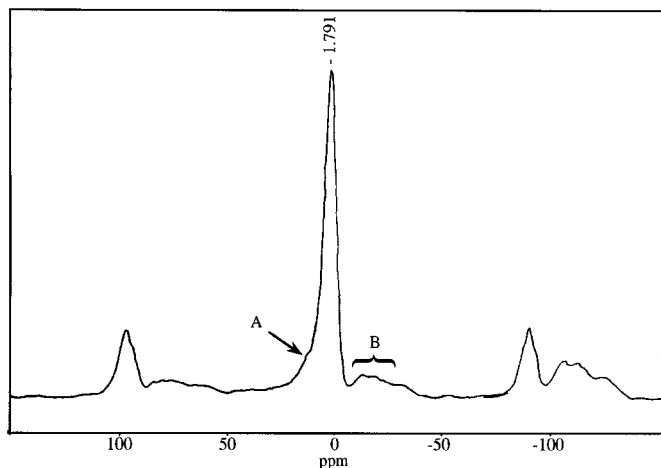


FIG. 6.  $^{11}\text{B}$  NMR spectra of  $\text{YBO}_3$  prepared by dry method.

spectra shows that boron atoms are differently coordinated. The spectrum of a sample with a 10%  $\text{Eu}^{3+}$  content is shown on Fig. 6. It is obvious that  $\text{YBO}_3$  contains mainly  $\text{BO}_4$  tetrahedra (20). However, the presence of a shoulder (denoted A) on Fig. 6 can be assigned to a different environment for boron which broadens the signal. The B peak could be consistent with boron in threefold coordination but the interpretation of the spectra is difficult in the range from  $-10$  to  $-50$  ppm.

### CONCLUSION

This new investigation of the crystal structure of  $\text{YBO}_3$  determined from X-ray single crystal data shows that yttrium atoms are eightfold coordinated. According to the spectroscopic results, two types of environments for yttrium are observed due to the delocalization of oxygen atoms. The boron atoms are in fourfold coordination with two kinds of anionic environments, one of which is very distorted.

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### REFERENCES

1. E. M. Levin, R. S. Roth, and J. B. Martin, *Am. Mineral.* **46**, 1030 (1961).
2. J. Weidelt and H. U. Bambauer, *Naturwissenschaften*, 342 (1968).
3. E. Antic-Fidancev, J. Aride, J. Chaminade, M. Lemaitre-Blaise, and P. Porcher, *J. Solid State Chem.* **97**, 74 (1992).
4. R. Bohlhoff, H. U. Bambauer, and W. Hoffman, *Z. Kristallogr.* **133**, 386 (1971).
5. R. E. Newnham, M. J. Redman, and R. P. Santoro, *J. Am. Ceram. Soc.* **46**, 253 (1963).
6. W. F. Bradley, D. L. Graf, and R. S. Roth, *Acta Crystallogr.* **20**, 283 (1966).
7. J. P. Laperches and P. Tarte, *Spectrochim. Acta.* **22**, 1201 (1966).
8. H. M. Kriz and P. J. Bray, *J. Chem. Phys.* **51**, 3642 (1969).
9. J. H. Denning and S. D. Ross, *Spectrochim. Acta Part A* **28**, 1775 (1972).
10. J. Hölsä, *Inorgan. Chim. Acta* **139**, 257 (1987).
11. J. P. Chaminade, A. Garcia, M. Pouchard, and C. Fouassier, *J. Crystal Growth* **99**, 799 (1990).
12. J. Cox and D. A. Keszler, *Acta Crystallogr. Sect. C* **50**, 1857 (1994).
13. A. Biedl, *Amer. Mineral.* **51** (1966).
14. G. Chadeyron, R. Mahiou, M. El-Ghozzi, A. Arbus, D. Zambon, and J. C. Cousseins, presented at ICL'96, Prague, 1996, to be published in *J. Luminescence*.
15. B. M. Wanklyn, *J. Crystal Growth* **37**, 334 (1977).
16. C. K. Fair, "MolEN — An Interaction Intelligent System for Crystal Structure Analysis," Enraf-Nonius, Delft, 1990.
17. D. T. Cromer, "International Tables For X-Ray Crystallography," Vol. 4, Kynoch Press, Birmingham, 1974.
18. C. E. Weir and R. A. Schroeder, *J. Res. Nat. Bur. Stand. A* **68**, 465 (1964).
19. V. I. Tsaryuk, V. D. Savchenko, V. F. Zolin, B. F. Dzhurinskii, G. V. Lysanova, and L. N. Margolin, *J. Appl. Spectrosc.* **59**, 895 (1993).
20. P. J. Bray, J. O. Edwards, J. G. O'Keeffe, V. F. Ross, and I. Tatsuzaki, *J. Chem. Phys.* **35**, 435 (1961).
21. H. M. Kriz, S. G. Bishop, and P. J. Bray, *J. Chem. Phys.* **49**, 557 (1968).
22. N. Walker and D. Stuart, *Acta Crystallogr. Sect. A* **39**, 158 (1983).